

# Supporting Information

Yim et al. 10.1073/pnas.1506939112

## SI Methods

**Calculation Details.** DFT-*vdW* calculations were performed using the Vienna *ab initio* simulation package (VASP, version 5.2.12) with the projector-augmented wave method and a cutoff of 500 eV for the plane wave expansion of the wave functions (42, 43). For the exchange-correlation functional, we have used the optB86b *vdW* functional implemented in VASP by Klimeš et al. (44), using the algorithm of Román-Pérez and Soler (45). A spin-restricted approach has been used as spin-polarization effects have been found to be negligible. Geometry optimizations were performed on a supercell structure using periodic boundary conditions. The (111) surface is modeled using a four-layer slab, with a CO adlayer on one side of the slab and a vacuum space equivalent to six ideal bulk metallic layers. We use the lattice constant of 3.946 Å, calculated theoretically for Pd, which is close to the experimental value of 3.891 Å. We calculated the average adsorption energies of CO molecules in three  $c(4 \times 2)$ -2CO unit cells with different site occupations of CO: In the first unit cell CO occupies both atop and bridge sites (TB), in the second unit cell CO occupies only bridge sites (BB), and in the last unit cell CO occupies both fcc and hcp hollow sites (HH). Note that all these unit cells correspond to the same CO coverage of 0.5 ML. We also tested the stability of these different  $c(4 \times 2)$ -2CO overlayers under anisotropic surface strain conditions. Tensile strain ( $\epsilon_s > 0$ ) is modeled by elongation along  $[\bar{1}01]$  (Fig. 5B) together with compression along  $[\bar{1}2\bar{1}]$  and vice versa for compressive strain ( $\epsilon_s < 0$ ). The ratio between the compressed strain applied along one direction and the tensile strain applied along the other (orthogonal) direction is set at 0.39, equal to the Poisson ratio of bulk Pd. During the calculations, the CO molecules and the two uppermost Pd substrate layers are allowed to relax, while the bottom two layers of Pd atoms are fixed in their bulk positions. All atomic coordinates of CO and Pd in the relaxed metal layers were optimized to a force of less than 0.01 eV/Å on each atom. Brillouin zone integration was performed using a Monkhorst-Pack grid of  $4 \times 4 \times 1$  for the  $c(4 \times 2)$  unit cells, and a Methfessel-Paxton smearing of 0.2 eV. Adsorption energies are calculated by subtracting the total energy of the adsorption system from that of the separated molecule and metal slab; the latter is under the same strain condition as the adsorption system.

**Analysis of  $\mu$ -NEXAFS Data.** Fig. 3B (*Inset*) shows an XPEEM image of a  $\text{TiO}_2(110)$  surface following Pd deposition. The image was collected with an electron kinetic energy of 0.5 eV and photon energy of 430 eV. As this photon energy is above the Pd  $M_{4,5}$  edges but still below the Ti  $L_{2,3}$  edges, the bright feature in the image with a measured diameter of about 1.5  $\mu\text{m}$  (circled) corresponds to a Pd nanoparticle, the dark background being the  $\text{TiO}_2(110)$  substrate (21). This surface was exposed to 120 L (1 L =  $1.33 \times 10^{-6}$  mbar.s) of CO *in situ* at room temperature. According to our STM experience, this should lead to  $\sim 0.5$  ML coverage of all of the Pd nanoparticle facets (1 ML =  $1.527 \times 10^{15}$  molecules. $\text{cm}^{-2}$ ).

Carbon K-edge ( $h\nu = 285$ – $293$  eV)  $\mu$ -NEXAFS measurements were recorded from this surface to determine the bond orien-

tation of CO on the Pd nanoparticle. This was achieved by recording a series of XPEEM images in the same surface region with an electron kinetic energy of 2 eV and photon energies ramping between 285 and 293 eV.

By assuming that the location of the Pd nanoparticle in the field-of-view in XPEEM does not change during the measurement, and by using a data mask, we extracted C K-edge X-ray absorption spectra from the Pd nanoparticle which are shown in Fig. 3A. The absorption peak at  $h\nu = 286.9$  eV corresponds to the  $\pi^*$  resonance of CO. This peak exhibits a strong dependence on the polarization angle  $\beta$ , shown in Fig. 3B (*Inset*).  $\beta$  represents the orientation of the electric vector of the incident photon beam. When  $\beta$  is  $0^\circ$ , the electric vector of the photon beam lies in the plane of incidence and corresponds to *p*-polarized light. Note that the angle of incidence  $\theta$  is always fixed at  $16^\circ$ . On the other hand, when  $\beta$  is  $90^\circ$ , the electric vector of the photon beam is perpendicular to the plane of incidence and corresponds to *s*-polarized light.

The areas of the  $\pi^*$  resonance at different polarization angles were obtained by fitting the  $\mu$ -NEXAFS curves in the vicinity of the  $\pi^*$  peak with a Gaussian function, as shown in Fig. 3B. To determine the bond angle of CO relative to the top facet of the Pd nanoparticle, we numerically fitted the experimental data points in Fig. 3B using Eqs. S1–S3, which are modified versions of those of Stöhr and Outka (31). This gives the polar bond angle of CO,  $\gamma$ , as  $21.4^\circ \pm 7.0^\circ$ , very close to the value obtained in our previous measurements using conventional NEXAFS (26). Although the nonzero bond angle seems to imply that the CO bonds with its molecular axis off the (111) top facet normal, the situation is rather more complicated. A significant proportion of CO molecules adsorbs on the side facets of the Pd nanoparticle and this must be accounted for. To do this, we simulated the  $\mu$ -NEXAFS measurements on the basis that CO bonds with its molecular axis parallel to the facet normal. In the simulation, we observed that as the contribution of the side facets to the X-ray absorption signal becomes larger, the normalized  $\pi^*$  peak area at  $\beta = 0^\circ$  increases to higher values and so the apparent bond angle moves further from the (111) top facet normal.

$$I_p = B \left[ \cos^2 \beta I_p^\parallel + \sin^2 \beta I_p^\perp \right], \quad \text{[S1]}$$

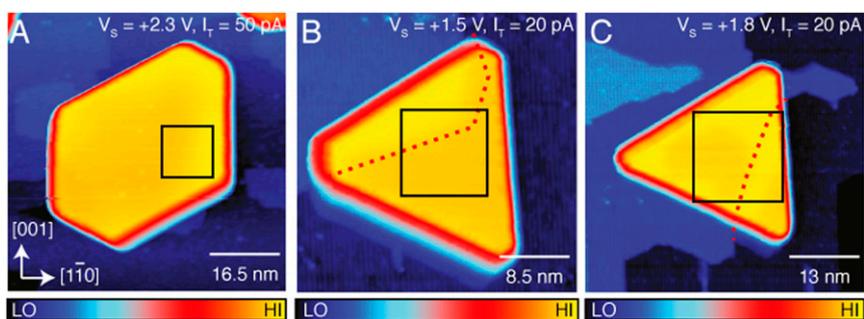
where

$$I_p^\parallel = \frac{2}{3} \left[ 1 - \frac{1}{4} (3 \cos^2 \theta - 1) (3 \cos^2 \gamma - 1) \right] \quad \text{[S2]}$$

and

$$I_p^\perp = \frac{1}{2} (1 + \cos^2 \gamma). \quad \text{[S3]}$$

In Eqs. S1–S3,  $B$  is an arbitrary constant,  $\beta$  is the polarization angle of the photon beam,  $\theta$  is the incident angle of the photon beam, and  $\gamma$  is the polar bond angle of CO relative to the surface normal of the (111) top facet.



**Fig. S1.** Topographical STM images of Pd nanocrystals with curved (111) top facets. (A) STM image ( $66 \times 66 \text{ nm}^2$ , 78 K) of the step-island shown in Fig. 2. A square marks the area on the top facet from which Fig. 2A was taken. The color bar has a range of 5.56 nm. (Scale bar, 16.5 nm.) (B) STM image ( $34 \times 34 \text{ nm}^2$ , 124 K) of the step-island shown in Fig. 4. The red dashed line roughly indicates the location of the substrate step edge that lies beneath the island and the square marks the area on the top facet where Fig. 4A was taken. The color bar has a range of 2.57 nm. (Scale bar, 8.5 nm.) (C) STM image ( $52 \times 52 \text{ nm}^2$ , 150 K) of the step-island shown in Fig. 6. The red dashed line roughly indicates the location of the substrate step edges that lies beneath the island and the square marks the area on the top facet where Fig. 6A was taken. The color bar has a range of 2.72 nm. (Scale bar, 13 nm.) It is clear that the island grows across the steps in A and is curved as a result. However, because of the greater height of the island, it is difficult to mark the locations of the  $\text{TiO}_2(110)$  step edges.